

Cu–O(4') est plus longue (2,50 Å). La présence d'un sixième atome d'oxygène O(2'') situé à une distance Cu–O(2'')=3,02 Å beaucoup trop grande du cuivre permet cependant de décrire cet environnement comme résultant de l'elongation d'un octaèdre oxygéné le long de l'axe quaternaire [ici le long de O(4')–O(2'')] par effet Jahn–Teller, phénomène classique dans les composés du cuivre (II).

Conclusions

La structure cristalline de la phase Cu₂V₂O₇ α est caractérisée par la présence de groupements anioniques [V₂O₇]⁴⁻ nettement individualisés. Ces groupements, dont le caractère covalent joue un rôle prépondérant, constituent des feuillets parallèles au plan yOz et reliés par les atomes de cuivre. La configuration même de ces anions permet de rattacher la variété basse température du pyrovanadate de cuivre au groupe structural de la thorveitite; cependant, leur disposition relative est originale et donne un caractère inédit à cette structure.

L'environnement du cuivre(II), quoique moins classique, a été signalé cependant dans la variété α du pyro-

phosphate de cuivre Cu₂P₂O₇ par Robertson & Calvo (1967).

L'étude structurale actuellement en cours de la variété Cu₂V₂O₇ β haute température permettra vraisemblablement d'expliquer le processus cristallographique de la transition $\alpha \rightleftharpoons \beta$.

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Crystal Structures of the Fluosilicate Hexahydrates of Cobalt, Nickel and Zinc*

BY SIDDHARTHA RAY, ALLAN ZALKIN AND DAVID H. TEMPLETON

Lawrence Berkeley Laboratory and Department of Chemistry, University of California, Berkeley, California 94720, U.S.A.

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Of seven fluosilicate hexahydrates supposedly isomorphous with NiSnCl₆.6H₂O, only CoSiF₆.6H₂O, NiSiF₆.6H₂O and ZnSiF₆.6H₂O have the typical space group ($R\bar{3}$) and cell size ($a=9\cdot366$ (2), $9\cdot313$ (3), $9\cdot363$ (3); $c=9\cdot730$ (4), $9\cdot623$ (2), $9\cdot690$ (5) Å, respectively, for the hexagonal cell). These three crystal structures were refined with single-crystal X-ray diffraction data obtained by counter methods to conventional R values of 0·033, 0·022 and 0·018. The structures are essentially the same, and there is disorder of F between two positions unrelated by any symmetry element and with unequal occupancy. With either of the disordered sites, the $M(H_2O)_6^{2+}$ ($M=Co$, Ni or Zn) and SiF₆²⁻ octahedra are stacked alternately along the threefold axis in approximately CsCl-type packing. The hydrogen atoms take part in hydrogen bonds with O–F distances ranging from 2·68 to 2·88 Å. Other bond distances are Co–O = 2·081, Ni–O = 2·047, Zn–O = 2·078, Si–F (average) = 1·676 Å.

Fluosilicate hexahydrates of several bivalent metals have been considered to be isomorphous with a large number of salts with the general formula MG₆LR₆, where M is a bivalent metal, G may be water or ammonia, L is a quadrivalent element like Si, Sn, Ti or Zr, and R may be Cl, F or CN. These crystals have a rhombohedrally distorted CsCl-type packing and similar cell dimensions, except for possible doubling of

axes (Hassel, 1927, 1931; Hassel & Richter-Salvesen, 1927).

Pauling (1930) determined a structure in space group $R\bar{3}$ for NiSnCl₆.6H₂O, and this result has been widely regarded as typical for the entire series. The oxygen and chlorine atoms are arranged in nearly regular octahedra with identical orientations and dimensions, and this structure must be regarded as approximate at best. In a neutron-diffraction study of FeSiF₆.6H₂O Hamilton (1962) deduced the space group to be $R\bar{3}m$. Disorder was postulated to explain this enhancement of

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symmetry because it gave better agreement with the data than did the alternative of twinning of crystals of $R\bar{3}$ symmetry. The structure is a superposition of two components, each of symmetry $R\bar{3}$, related to each other by a mirror. The component structure resembles that determined by Pauling except for significant differences in the dimensions and orientations of the two kinds of octahedra.

A phase transition occurs near 230°K in $MnSiF_6 \cdot 6H_2O$ (Tsujikawa & Couture, 1955) and near 250°K in $CoSiF_6 \cdot 6H_2O$ (Ray, 1964; Majumdar & Datta, 1965). While studying these transitions we became interested in the room-temperature structures of this series of salts and learned that they are remarkably diverse in spite of the close isomorphism in the sense of similar axial ratios. We have examined crystals of fluosilicate hexahydrates of Mg, Mn, Co, Ni, Cu and Zn with the result that all are different from the Fe salt and only the Co, Ni and Zn salts have the typical cell dimensions and space group. The Cu salt also exhibits space group $R\bar{3}$ but with a cell four times as large. The Mn compound has a primitive hexagonal cell rather than a rhombohedral one, a fact also reported by Kodera, Torii & Osaki (1972). In our work at 23°C we observed crystals of the Mg salt to be of lower symmetry than trigonal, but twinned. Syoyami & Osaki (1972) discovered a transition at about 25° in this salt, with space group $R\bar{3}m$ above the transition. They determined the structure of the low-temperature form to be monoclinic, space group $P2_1/c$.

The present paper describes the structures of the Co, Ni and Zn salts; they differ significantly from the structure reported by Pauling (1930) for $NiSnCl_6 \cdot 6H_2O$.

Experimental

The salts $CoSiF_6 \cdot 6H_2O$, $NiSiF_6 \cdot 6H_2O$ and $ZnSiF_6 \cdot 6H_2O$ were prepared by dissolving respectively cobalt carbonate, nickel carbonate and metallic zinc in fluosilicic acid and slowly evaporating the filtered solutions. Crystals were selected from crops recrystallized from aqueous solution by slow evaporation.

The Co crystal selected for study was a hexagonal (almost triangular) prism of length 0.13 mm and maximum thickness 0.10 mm. The cell dimensions were derived from measurements of 2θ for $h00$, $hh0$ and $00l$ reflections with a manual General Electric goniostat, using Mo $K\alpha$ radiation ($\lambda=0.70926 \text{ \AA}$ for α_1). The α doublet was resolved in each case. Intensity data were collected with Zr-filtered Mo radiation and a card-controlled General Electric automatic diffractometer which was equipped with a scintillation counter and pulse-height discriminator. The $\theta-2\theta$ scan technique was used with 2θ scanned at the rate of 1° min^{-1} , starting 0.6° below the α_1 peak and ending 0.6° beyond the α_2 peak, for all reflections with l zero or positive and $2\theta < 70^\circ$. Background was counted for 10 s each at 0.5° below the start and 0.5° beyond the end of each scan. Four reflections were checked at regular intervals,

and no systematic variation was observed. The measurements included 2666 reflections of which 721 were independent; 60 of these had observed intensity less than the estimated standard deviation.

The Ni crystal was of rhombohedral shape with height ~ 0.2 mm and maximum thickness ~ 0.2 mm. It showed all six faces of the form $\{01\bar{1}2\}$, three faces of the form $\{10\bar{1}1\}$, and four of the form $\{11\bar{2}0\}$. A computer-controlled Picker four-circle diffractometer with a scintillation counter and pulse-height discriminator was used for the diffraction experiments with Cu $K\alpha$ radiation and an oriented-graphite monochromator. Cell dimensions were derived by least squares from measurements of 12 high-angle reflections ($\lambda=1.54051 \text{ \AA}$ for α_1). Intensities were scanned as described above. The whole sphere of reciprocal space out to $2\theta=162^\circ$, except for certain blind regions, was explored. Above 128° , the χ circle was in the parallel position. Standard reflections were checked periodically, and no systematic variation was observed. 1873 reflections were measured, of which 298 were independent. Only 4 of these had intensity less than the standard deviation.

The Zn crystal was a regular hexagonal prism of height 0.10 mm and thickness 0.10 mm. Cell dimensions were derived by least squares from measurements of 24 high-angle reflections with the manual goniostat, the α -doublet being resolved in each case, and using Cu $K\alpha$ radiation. Intensities were measured with the Picker diffractometer as described above, except that monochromated Mo $K\alpha$ radiation was used. The whole sphere out to $2\theta=70^\circ$ was explored. No significant variation of standard reflections was observed. Intensities of 726 independent reflections, of which 92 were less than the standard deviation, were obtained from 4336 reflections.

Linear absorption coefficients of the Co and Zn salts are respectively 20.4 and 27.8 cm^{-1} for Mo $K\alpha$ radiation. An estimation of absorption correction factors for the first case indicated less than 2% difference between maximum and minimum values, and the other case is comparable. No corrections were made for these two crystals. For the Ni compound, $\mu=50.3 \text{ cm}^{-1}$ for Cu $K\alpha$ radiation, and a correction was made by numerical integration. The correction factors ranged from 1.40 to 2.19, and they significantly improved the agreement for cases in which the same reflection was measured in both the 'bisecting' and 'parallel' positions of the χ circle.

Intensities were corrected for background, converted to structure-factor magnitudes averaged for measurements of equivalent reflections, and assigned standard deviations which included contributions for counting statistics, excessive scatter, and systematic effects proportional to intensity as described by Fischer, Templeton & Zalkin (1970). The factor p for systematic effects was chosen as 0.05, 0.06 and 0.025 respectively for the Co, Ni and Zn salts.

A CDC-6600 computer was used for the calculations. Refinement by least squares was carried out with a

full-matrix program which minimizes $R^2 = \sum w(\Delta F)^2 / \sum w|F_o|^2$, where $w = [\sigma(F)]^{-2}$. Temperature factors were of the form $\exp(-B\lambda^{-2} \sin^2 \theta)$ if isotropic or $\exp(-\sum B_{ij} b_i b_j h_i h_j / 4)$ if anisotropic, where b_i is the magnitude of the i th reciprocal axis. Neutral-atom form factors of Cromer & Waber (1965) were used for atoms other than hydrogen, for which the values of Stewart, Davidson & Simpson (1965) were used. Anomalous dispersion corrections (Cromer, 1965) were, for Mo $K\alpha$, Co: 0.37 + i 1.06, Zn: 0.33 + i 1.54, Si: 0.09 + i 0.09; for Cu $K\alpha$, Ni: -3.20 + i 0.67, Si: 0.23 + i 0.36.

Crystal data

$\text{CoSiF}_6 \cdot 6\text{H}_2\text{O}$: reddish-pink hexagonal prisms.

Hexagonal cell: $a = 9.366$ (2), $c = 9.730$ (4) Å, $Z = 3$.

Rhombohedral cell: $a = 6.306$ (3) Å, $\alpha = 95^\circ 55'$ (4)', $Z = 1$.

Space group: $R\bar{3}$; $D_m = 2.07$, $D_x = 2.08$ g cm⁻³.

$\text{NiSiF}_6 \cdot 6\text{H}_2\text{O}$: apple-green rhombohedra.

Hexagonal cell: $a = 9.313$ (3), $c = 9.623$ (2) Å, $Z = 3$.

Rhombohedral cell: $a = 6.261$ (2) Å, $\alpha = 96^\circ 6'$ (5)', $Z = 1$.

Space group: $R\bar{3}$; $D_m = 2.12$, $D_x = 2.13$ g cm⁻³.

$\text{ZnSiF}_6 \cdot 6\text{H}_2\text{O}$: colorless hexagonal prisms.

Hexagonal cell: $a = 9.363$ (3), $c = 9.690$ (5) Å, $Z = 3$.

Rhombohedral cell: $a = 6.297$ (3) Å, $\alpha = 96^\circ 3'$ (6)', $Z = 1$.

Space group: $R\bar{3}$; $D_m = 2.12$, $D_x = 2.13$ g cm⁻³.

Atoms were assigned coordinates similar to those of one of the lower symmetry ($R\bar{3}$) components of $\text{FeSiF}_6 \cdot 6\text{H}_2\text{O}$ (Hamilton, 1962). With isotropic temperature factors, least-squares refinement reduced the conventional $R = \sum |\Delta F| / \sum |F_o|$ to 0.16. With anisotropic temperature factors, R was reduced to 0.11, but B_{11} and B_{22} for F were unusually high. A Fourier synthesis of ΔF showed two prominent but unequal peaks, one on each side (in the same xy plane) of the assumed position of F. The only satisfactory explanation of this was the assumption of a disordered model. Fractional atoms, referred to as F(1) and F(2), were assigned to these two positions with occupation factors adding to unity. With equal occupancy, R was reduced to 0.07; with $F(1)/F(2) = \frac{1}{2}$, $R = 0.05$. Serious discrepancies remained between F_o and F_c for the $00l$ reflections. A ΔF map at this stage showed clearly the positions of the two hydrogen atoms. Introduction of these atoms with isotropic temperature factors removed the above discrepancies and reduced R to 0.033. Trial of various values of the occupation factors yielded the best fit with $F(1)/F(2) = 0.43/0.57$: $R = 0.033$ [excluding data where $I < \sigma(I)$], $R = 0.038$ (all data), $R_2 = 0.038$, estimated standard deviation of an observation of unit weight = 1.03. In the last cycle no parameter shifted more than 3% of its standard deviation.

In the case of $\text{NiSiF}_6 \cdot 6\text{H}_2\text{O}$ we introduced an empirical extinction correction: $F'_o = F_o(1 + 1.3 \times 10^{-7} I)$, where F'_o is the corrected structure factor. The correction factor was 1.43 for the strongest reflection, 110, and exceeded 1.13 for only three other reflections. With occupancy factors $F(1)/F(2) = 0.33/0.67$, $R = 0.022$ (with or without the four zero-weighted data), $R_2 = 0.040$, and the standard deviation of an observa-

Determination of the structure

The structure was determined in much the same way in each case, but we give more details for the Co salt.

Table 1. Coordinates and thermal parameters, with estimated standard deviations in parentheses

	<i>x</i>	<i>y</i>	<i>z</i>	B_{11} or B	B_{22}	B_{33}	B_{12}	B_{13}	B_{23}
<i>(a)</i> $\text{CoSiF}_6 \cdot 6\text{H}_2\text{O}$									
Co	0	0	0	2.00 (1)	2.00	1.45 (2)	1.00	0	0
Si	0	0	$\frac{1}{2}$	1.88 (2)	1.88	1.35 (3)	0.94	0	0
F(1)	0.1536 (4)	0.1356 (4)	0.4001 (3)	3.1 (1)	4.0 (1)	2.16 (8)	0.5 (1)	0.46 (8)	0.5 (1)
F(2)	0.1673 (3)	0.0679 (3)	0.3997 (2)	2.30 (7)	4.9 (1)	2.58 (7)	1.42 (8)	0.60 (5)	0.23 (9)
O	0.1745 (2)	0.1860 (2)	0.1247 (1)	3.84 (6)	3.52 (4)	2.26 (4)	0.94 (4)	-0.66 (4)	-0.28 (3)
H(1)	0.187 (4)	0.172 (4)	0.212 (3)	4.6 (6)					
H(2)	0.191 (3)	0.278 (4)	0.114 (2)	4.3 (6)					
<i>(b)</i> $\text{NiSiF}_6 \cdot 6\text{H}_2\text{O}$									
Ni	0	0	0	1.79 (3)	1.79	1.34 (4)	0.895	0	0
Si	0	0	$\frac{1}{2}$	1.71 (3)	1.71	1.22 (5)	0.855	0	0
F(1)	0.1610 (8)	0.1258 (8)	0.3999 (7)	2.9 (2)	4.4 (3)	2.0 (2)	-0.6 (2)	0.6 (1)	0.1 (2)
F(2)	0.1681 (3)	0.0647 (3)	0.3981 (3)	2.20 (7)	3.8 (1)	2.46 (9)	1.22 (9)	0.49 (5)	-0.1 (1)
O	0.1772 (2)	0.1830 (1)	0.1218 (1)	3.02 (5)	2.17 (5)	2.20 (6)	0.88 (4)	-0.48 (4)	-0.15 (4)
H(1)	0.181 (3)	0.164 (3)	0.209 (3)	4.2 (5)					
H(2)	0.184 (3)	0.262 (3)	0.116 (2)	4.3 (6)					
<i>(c)</i> $\text{ZnSiF}_6 \cdot 6\text{H}_2\text{O}$									
Zn	0	0	0	1.968 (9)	1.968	1.49 (1)	0.984	0	0
Si	0	0	$\frac{1}{2}$	1.68 (1)	1.68	1.23 (2)	0.84	0	0
F(1)	0.1555 (4)	0.1326 (4)	0.4001 (3)	2.8 (1)	3.8 (1)	2.08 (8)	0.2 (1)	0.52 (7)	0.53 (9)
F(2)	0.1676 (2)	0.0685 (2)	0.3992 (2)	2.25 (4)	4.56 (8)	2.36 (4)	1.33 (5)	0.59 (3)	0.14 (5)
O	0.1766 (1)	0.1853 (1)	0.1240 (1)	3.47 (4)	2.35 (3)	2.18 (3)	0.95 (3)	-0.62 (2)	-0.21 (2)
H(1)	0.184 (2)	0.169 (2)	0.202 (2)	4.0 (4)					
H(2)	0.192 (2)	0.274 (2)	0.117 (2)	4.3 (4)					

Table 2. Observed and calculated structure factors

Data where $I < \sigma(I)$ are marked with an asterisk.

TABLE OF OBSERVED AND CALCULATED STRUCTURE FACTORS FOR COBALT FLUOSILICATE HEXAHYDRATE

EEA10-8-81 = 3294

TABLE OF OBSERVED AND CALCULATED STRUCTURE FACTORS FOR ZINC FLUOSILICATE HEXAHYDRATE

tion of unit weight = 1.33. In the last cycle, no parameter shifted more than 1% of its standard deviation. Slight variation of the occupancy ratio did not result in any significant improvement.

In the otherwise straightforward refinement of the structure of $\text{ZnSiF}_6 \cdot 6\text{H}_2\text{O}$ we were confronted with the interesting fact that in this centric space group ($R\bar{3}$) different values of the coordinates must be assigned to identical structures if opposite choices are made for the positive direction of the unique axis. By chance, the same choice was made for the Co and Ni crystals, but the opposite choice for the Zn crystal. Thus the Zn structure was determined without the knowledge that it was so nearly the same as the others. The coordinates and indices had to be transformed by the matrix

$$\begin{pmatrix} 0 & 1 & 0 \\ 1 & 0 & 0 \\ 0 & 0 & -1 \end{pmatrix}$$

to yield the results reported here. No meaningful improvement was gained when the occupancy ratio was changed from $F(1)/F(2) = 0.33/0.67$, and with this ratio $R = 0.018$ excluding reflections with $I < \sigma(I)$; $R = 0.023$ (all reflections), $R_2 = 0.021$; standard deviation of an observation of unit weight = 1.07. In the last cycle no parameter shifted by more than 4% of its standard deviation.

The final parameters are given in Table 1. Observed and calculated structure factors are listed in Table 2. Some interatomic distances and angles appear in Tables 3, 4 and 5.

Table 3. Interatomic distances (Å)

Values in parentheses are corrected for thermal motion, assuming that the lighter atom rides on the heavier atom. Distances involving disordered positions F(1) and F(2) are designated by (1) and (2) respectively.

(a) $\text{CoSiF}_6 \cdot 6\text{H}_2\text{O}$			
Co—O	2.081 ± 0.001	(2.092)	
Si—F	(1) 1.674 ± 0.001	(1.695)	
	(2) 1.678 ± 0.002	(1.697)	
O—H(1)	0.874 ± 0.026	(0.888)	
O—H(2)	0.805 ± 0.032	(0.822)	
(b) $\text{NiSiF}_6 \cdot 6\text{H}_2\text{O}$			
Ni—O	2.047 ± 0.001	(2.055)	
Si—F	(1) 1.671 ± 0.006	(1.701)	
	(2) 1.683 ± 0.003	(1.698)	
O—H(1)	0.862 ± 0.029	(0.881)	
O—H(2)	0.709 ± 0.026	(0.737)	
(c) $\text{ZnSiF}_6 \cdot 6\text{H}_2\text{O}$			
Zn—O	2.078 ± 0.001	(2.088)	
Si—F	(1) 1.671 ± 0.003	(1.693)	
	(2) 1.680 ± 0.003	(1.698)	
O—H(1)	0.778 ± 0.020	(0.790)	
O—H(2)	0.775 ± 0.020	(0.797)	

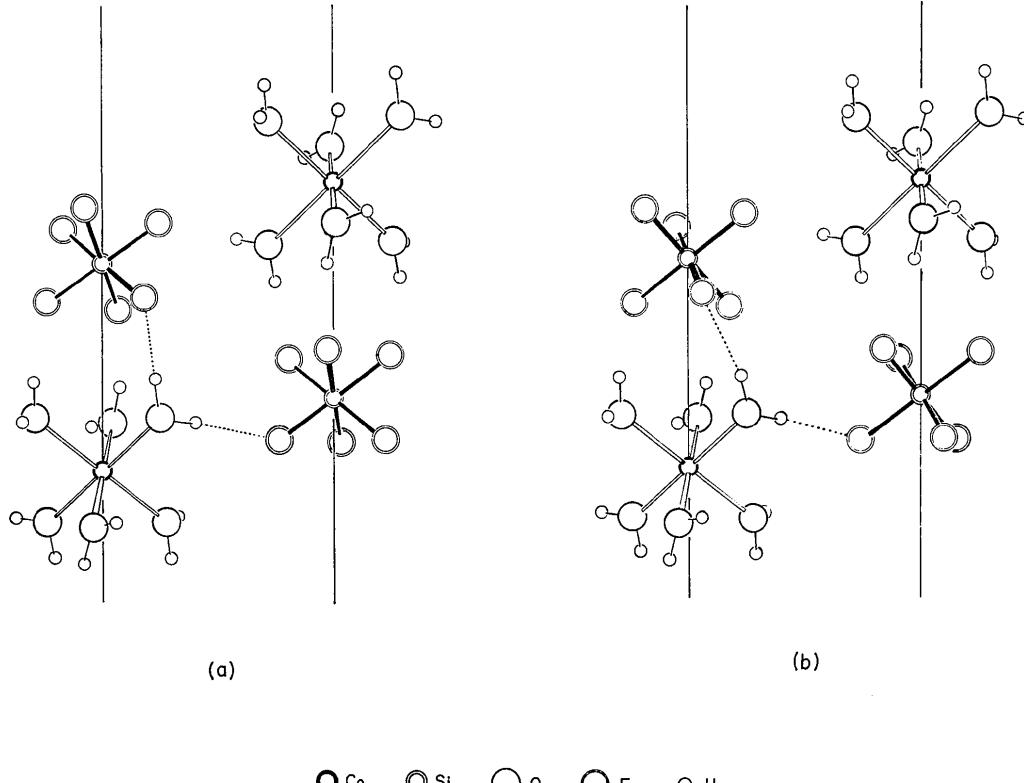


Fig. 1. Two neighboring columns in the structure of $\text{CoSiF}_6 \cdot 6\text{H}_2\text{O}$. Fluorine atoms are shown in (a) according to the F(1) positions and in (b) according to F(2).

Table 4. Distances involved in hydrogen bonds (Å)

F_{\parallel} and F_{\perp} refer respectively to bonds approximately parallel and perpendicular to the threefold axis.

(a) $\text{CoSiF}_6 \cdot 6\text{H}_2\text{O}$		
$\text{O}—\text{F}_{\parallel}$	(1) 2.711 ± 0.003	(2) 2.888 ± 0.003
$\text{H}(1)—\text{F}_{\parallel}$	(1) 1.861 ± 0.026	(2) 2.037 ± 0.029
$\text{O}—\text{F}_{\perp}$	(1) 2.721 ± 0.004	(2) 2.759 ± 0.003
$\text{H}(2)—\text{F}_{\perp}$	(1) 1.961 ± 0.031	(2) 1.957 ± 0.033
(b) $\text{NiSiF}_6 \cdot 6\text{H}_2\text{O}$		
$\text{O}—\text{F}_{\parallel}$	(1) 2.718 ± 0.007	(2) 2.863 ± 0.004
$\text{H}(1)—\text{F}_{\parallel}$	(1) 1.862 ± 0.031	(2) 2.017 ± 0.031
$\text{O}—\text{F}_{\perp}$	(1) 2.683 ± 0.007	(2) 2.782 ± 0.003
$\text{H}(2)—\text{F}_{\perp}$	(1) 2.021 ± 0.028	(2) 2.082 ± 0.027
(c) $\text{ZnSiF}_6 \cdot 6\text{H}_2\text{O}$		
$\text{O}—\text{F}_{\parallel}$	(1) 2.710 ± 0.004	(2) 2.868 ± 0.002
$\text{H}(1)—\text{F}_{\parallel}$	(1) 1.948 ± 0.020	(2) 2.104 ± 0.021
$\text{O}—\text{F}_{\perp}$	(1) 2.713 ± 0.003	(2) 2.766 ± 0.002
$\text{H}(2)—\text{F}_{\perp}$	(1) 1.979 ± 0.020	(2) 1.995 ± 0.021

Table 5. Bond angles ($^{\circ}$)

(a) $\text{CoSiF}_6 \cdot 6\text{H}_2\text{O}$		
$\text{O}—\text{Co}—\text{O}$		89.4 ± 0.1
$\text{O}—\text{Co}—\text{O}'$		90.6 ± 0.1
$\text{F}—\text{Si}—\text{F}$	(1) 89.6 ± 0.2	(2) 89.6 ± 0.2
$\text{F}—\text{Si}—\text{F}'$	(1) 90.4 ± 0.2	(2) 90.4 ± 0.2
$\text{H}(1)—\text{O}—\text{H}(2)$		108.7 ± 3.0
$\text{O}—\text{H}(1) \cdots \text{F}$	(1) 163.7 ± 1.0	(2) 162.7 ± 1.0
$\text{O}—\text{H}(2) \cdots \text{F}$	(1) 157.0 ± 1.0	(2) 174.0 ± 0.8
$\text{F} \cdots \text{O} \cdots \text{F}$	(1) 110.7 ± 0.1	(2) 123.5 ± 0.1
(b) $\text{NiSiF}_6 \cdot 6\text{H}_2\text{O}$		
$\text{O}—\text{Ni}—\text{O}$		90.5 ± 0.1
$\text{O}—\text{Ni}—\text{O}'$		89.5 ± 0.1
$\text{F}—\text{Si}—\text{F}$	(1) 90.1 ± 0.4	(2) 89.4 ± 0.2
$\text{F}—\text{Si}—\text{F}'$	(1) 89.9 ± 0.4	(2) 90.6 ± 0.2
$\text{H}(1)—\text{O}—\text{H}(2)$		107.4 ± 3.0
$\text{O}—\text{H}(1) \cdots \text{F}$	(1) 171.9 ± 1.0	(2) 166.4 ± 0.7
$\text{O}—\text{H}(2) \cdots \text{F}$	(1) 155.6 ± 1.0	(2) 169.5 ± 0.7
$\text{F} \cdots \text{O} \cdots \text{F}$	(1) 111.5 ± 0.2	(2) 123.1 ± 0.1
(c) $\text{ZnSiF}_6 \cdot 6\text{H}_2\text{O}$		
$\text{O}—\text{Zn}—\text{O}$		89.9 ± 0.1
$\text{O}—\text{Zn}—\text{O}'$		90.1 ± 0.1
$\text{F}—\text{Si}—\text{F}$	(1) 89.8 ± 0.2	(2) 89.6 ± 0.1
$\text{F}—\text{Si}—\text{F}'$	(1) 90.2 ± 0.2	(2) 90.4 ± 0.1
$\text{H}(1)—\text{O}—\text{H}(2)$		108.1 ± 2.6
$\text{O}—\text{H}(1) \cdots \text{F}$	(1) 166.3 ± 0.7	(2) 167.0 ± 0.5
$\text{O}—\text{H}(2) \cdots \text{F}$	(1) 157.8 ± 0.7	(2) 170.8 ± 0.6
$\text{F} \cdots \text{O} \cdots \text{F}$	(1) 110.9 ± 0.1	(2) 123.2 ± 0.1

Discussion

Except for minor numerical details the three salts have the same crystal structure. An unusual feature of this structure, for which we have no satisfactory explanation, is the disorder of F between two sites, unrelated by any symmetry element and with unequal occupancy factors. The distances between alternate sites are 0.71, 0.60 and 0.66 Å, respectively, in the Co, Ni and Zn salts. We assume that all the F atoms in a particular SiF_6^{2-} ion are in the same type of site because otherwise there would be unsatisfactory F-F distances. According to this model, the disorder involves two orienta-

tions of the fluosilicate ion about 30° apart with respect to rotation about the threefold axis. The F atoms are at the corners of regular or nearly regular octa-

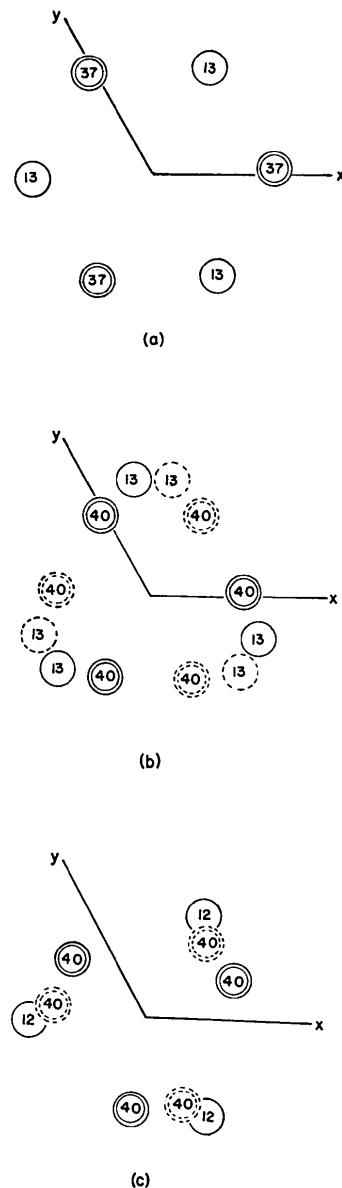


Fig. 2. Relative orientation of oxygen atoms on the top face of a water octahedron and halogen atoms on the bottom face of the adjacent octahedron in (a) $\text{NiSnCl}_6 \cdot 6\text{H}_2\text{O}$ (Pauling, 1930), (b) $\text{FeSiF}_6 \cdot 6\text{H}_2\text{O}$ (Hamilton, 1962) and (c) $\text{CoSiF}_6 \cdot 6\text{H}_2\text{O}$ (this study). The numbers indicate values of the z coordinates ($\times 100$), and broken circles indicate the alternate sites according to the disorder models.

○, ○ : Oxygen

○, ○ : Halogen

hedra; for the F(1) sites in the Ni and Zn compounds the octahedra are regular within the experimental accuracy, while in the other cases there is a slight elongation in the direction of the threefold axis of two to four standard deviations. All six results are within about one standard deviation of an average shape with $F-Si-F=89.7^\circ$, as is also the result found by Hamilton (1962) for the Fe salt.

The water molecules are also at the corners of nearly regular octahedra, but these octahedra are not so consistent in shape. In the Zn salt the octahedron is regular, while for Co it is elongated and for Ni it is compressed, but by amounts which change bond angles only about half a degree. Hamilton (1962) observed a greater elongation [$O-Fe-O=88.6(3)^\circ$] in the Fe salt.

These octahedra are packed according to a rhombohedrally distorted CsCl-type structure. The densest packing is in columns of alternate cations and anions along the threefold axis (Fig. 1). Three hydrogen bonds connect each pair of adjacent octahedra in a column, while one hydrogen bond connects an octahedron to each of its six neighboring octahedra in other columns. This topology of hydrogen bonding is the same for either F site and is also the same as that which Hamilton (1962) deduced for each component of his structure. It would not exist in the Pauling (1930) structure because in that arrangement each O would have two equidistant F neighbors in the same column (Fig. 2). This latter arrangement permits closer packing of spherical atoms, but provides too many acceptors for hydrogen bonds. Any rotation of one kind of octahedron relative to the other makes one neighbor closer than the other, with a rotation of 60° giving the maximum difference. In the $FeSiF_6 \cdot 6H_2O$ structure, the oxygen atoms are twisted about 40° from the Pauling

(1930) structure, in each component of the disorder model. In the present structures it is the fluorine atoms which are twisted, F(1) about 50° and F(2) about 20° .

Note added in proof: – Lynton & Siew (1973) describe the structure of the cobalt salt with fluorine in a single site with large and very anisotropic thermal motion. Their measurements ended at $2\theta=40^\circ$ with Mo K α radiation, while ours extended to 70° . This model is similar to the one which we found to be inadequate to fit our more extensive data.

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